## X-ray diffraction Analysis <br> Using fullprof suit

X-ray Tube

Presented By Geeta Ray


## X-Ray Diffraction

## What is diffraction?

- incident radiation (e.g., light, X-rays) scatters as it passes through a finely spaced periodic array (e.g., grating, crystal lattice) polychromatic (white) light

- where beams of scattered radiation emerge from slit "in phase", constructive interference produces "diffraction maxima"
- position and intensity of maxima depends on spacing of array and integral number of $\lambda$ contributing to signal ( $\mathrm{n} \lambda$ )


## WHY X-RAYS?

-For electromagnetic radiation to be diffracted the spacing in the grating should be of the same order as the wavelength

- In crystals the typical interatomic spacing ~ 2-3 Å so the suitable radiation is X-rays
- Hence, X-rays can be used for the study of crystal structures


Table 8.1 Characteristic Wavelengths ( $\AA$ ) of Metals Commonly Used as Targets in X-Ray Tubes ${ }^{\text {a }}$

|  | Metal |  |  |  |  |  |  |  |  |  |
| :--- | :--- | :--- | :--- | :--- | :--- | :---: | :---: | :---: | :---: | :---: |
|  | Mo |  |  |  |  |  | $\mathbf{C u}$ | $\mathbf{C o}$ | $\mathbf{F e}$ | $\mathbf{C r}$ |
| $K \beta$ | 0.63225 | 1.38217 | 1.62073 | 1.75653 | 2.08479 |  |  |  |  |  |
| $K \alpha_{1}$ | 0.70926 | 1.54051 | 1.78892 | 1.93597 | 2.28962 |  |  |  |  |  |
| $K \alpha_{2}$ | 0.71354 | 1.54433 | 1.79279 | 1.93991 | 2.29351 |  |  |  |  |  |
| $K \bar{\alpha}$ | 0.7107 | 1.5418 | 1.7902 | 1.9373 | 2.2909 |  |  |  |  |  |

${ }^{\text {a }} K \bar{\alpha}$ is the weighted average of $K \alpha_{1}$ and $K \alpha_{2}$.

## X-Ray Diffraction

What is X-ray diffraction (XRD) crystallography?

- periodic atomic arrays in crystal lattice act like 3-D diffraction gratings - for practical purposes, diffraction can be treated like reflection from multiple equivalent lattice planes ( $h k l$ )


> sharp peaks
broad peaks


Liquid or Glass
diffuse, continuous
spectrum

## BRAGG VIEW OF DIFFRACTION

X-rays that hit the crystal are elastically scattered by the sets of (hkl) planes
The path difference for rays 1 and 2

$$
\Delta(1-2)=2 d_{h k l} \sin \theta
$$ equals to the length of two blue lines:



## $\times-$ d.cpp ( $\sim$ - gedit

File Edit View Search Tools Documents Help

```
T}\mathrm{ Open * Save & Undo
    d.cpp *
    1 #include<iostream>
    2 #include<math.h>
    3 using namespace std;
    main()
    5 {
    float twotheta,d,s,l,pi;
    7 cout<<"enter two theta value\n";
    8 cin>>twotheta;
    9 pi=4*atan(1);
10 l=twotheta*pi/360;
1 1
12 s=2*sin(l);
13 d=1.54/s;
14
15 cout<<"d="<< d<<"\n";
16 }
N
```

$\infty-\square$ crystallab@ubuntu: ~
File Edit View Search Terminal Help
crystallab@ubuntu: $\sim \$ \mathrm{~g}++\mathrm{d}$. cpp crystallab@ubuntu: ~\$ . /a. out enter two theta value
9.608
$d=9.19431$
crystallab@ubuntu: ~\$ . /a. out enter two theta value
10.628
$d=8.31409$
crystallabgubuntu: ~\$ . /a.out enter two theta value
12. 563
$d=7.03753$
crystallabgubuntu: $-\$$. /a.out enter two theta value
15.935
$\mathrm{d}=5.5551$
crystallabgubuntu: $-\$$. /a. out enter two theta value
17.609
$\mathrm{d}=5.03059$
crystallabgubuntu: $-\$$. /a.out
enter two theta value
18.192
$\mathrm{d}=4.87067$
crystallab@ubuntu: - \$ . /a.out
enter two theta value
19.399
$d=4.57025$
crystallab@ubuntu: -\$ . /a, out
enter two theta value
19.743
$d=4.49139$
crystallab@ubuntu: $-\$$. /a. out
enter two theta value

| System | Lattice Parameters |
| :--- | :--- |
| Triclinic | $a \neq b \neq c$ <br> $\alpha \neq \beta \neq \gamma \neq 90$ |
| Monoclinic | $a \neq b \neq c$ <br> $\alpha=\gamma=90, \beta \neq 90$ |
| Orthorhombic | $a \neq b \neq c$ <br> $\alpha=\beta=\gamma=90$ |
| Tetragonal | $a=b \neq c$ <br> $\alpha=\beta=\gamma=90$ |
| Hexagonal | $a=b \neq c$ <br> $\alpha=\beta=90, \gamma=120$ |
| Rhombohedral | $a=b=c$ <br> (Trigonal) |
| Cubic | $a=b=\gamma \neq 90$ <br>  |

Cubic:

$$
\frac{1}{d^{2}}=\frac{h^{2}+k^{2}+l^{2}}{a^{2}}
$$

Tetragonal: $\quad \frac{1}{d^{2}}=\frac{h^{2}+k^{2}}{a^{2}}+\frac{l^{2}}{c^{2}}$
Hexagonal: $\quad \frac{1}{d^{2}}=\frac{4}{3}\left(\frac{h^{2}+h k+k^{2}}{a^{2}}\right)+\frac{l^{2}}{c^{2}}$
Rhombohedral:

$$
\frac{1}{d^{2}}=\frac{\left(h^{2}+k^{2}+l^{2}\right) \sin ^{2} \alpha+2(h k+k l+h l)\left(\cos ^{2} \alpha-\cos \alpha\right)}{a^{2}\left(1-3 \cos ^{2} \alpha+2 \cos ^{3} \alpha\right)}
$$

Orthorhombic:
$\frac{1}{d^{2}}=\frac{h^{2}}{a^{2}}+\frac{k^{2}}{b^{2}}+\frac{l^{2}}{c^{2}}$
Monoclinic: $\quad \frac{1}{d^{2}}=\frac{1}{\sin ^{2} \beta}\left(\frac{h^{2}}{a^{2}}+\frac{k^{2} \sin ^{2} \beta}{b^{2}}+\frac{l^{2}}{c^{2}}-\frac{2 h l \cos \beta}{a c}\right)$
Triclinic: $\quad \frac{1}{d^{2}}=\frac{1}{V^{2}}\left(S_{11} h^{2}+S_{22} k^{2}+S_{33} l^{2}+2 S_{12} h k+2 S_{23} k l+2 S_{13} h l\right)$
In the equation for triclinic crystals,

$$
\begin{aligned}
V & =\text { volume of unit cell (see below) } \\
S_{11} & =b^{2} c^{2} \sin ^{2} \alpha \\
S_{22} & =a^{2} c^{2} \sin ^{2} \beta \\
S_{33} & =a^{2} b^{2} \sin ^{2} \gamma \\
S_{12} & =a b c^{2}(\cos \alpha \cos \beta-\cos \gamma) \\
S_{23} & =a^{2} b c(\cos \beta \cos \gamma-\cos \alpha) \\
S_{13} & =a b^{2} c(\cos \gamma \cos \alpha-\cos \beta)
\end{aligned}
$$

Possible space groups
For monoclinic system
Pm, P2/m

P2 ${ }^{1}, \mathrm{P}^{1}{ }^{1 / m}$
hkl: none h0l: none
$0 \mathrm{kO}: \mathrm{k}=2 \mathrm{n}+1$
hkl: none
hOl: $\mathrm{I}=2 \mathrm{n}+1$
0k0: none
P2 ${ }^{1 / c}$

C2, Cm, C2/m
Systematic absences
hkl: none h0l: none 0kO: none

Pc, P2/c
hkl: none
h0l: $\mathrm{I}=2 \mathrm{n}+1$
0 kO : $\mathrm{k}=2 \mathrm{n}+1$
C2, Cm, C2/m
hOl: $(h=2 n+1)$
OkO: $(\mathrm{k}=2 \mathrm{n}+1)$

## MILLER PLANES

Atoms form periodically arranged planes
Any set of planes is characterized by:
(1) their orientation in the crystal (hkl) - Miller indices
(2) their $d$-spacing $\left(d_{h k}\right)$ - distance between the planes
$h, k$, l correspond to the number of segments in which the $a, b, c$ axes, respectively, are cut by the set of planes

On average, the higher (hkl),
the closer is the interplanar distance, $d_{h k l}$



## Intensity of the Scattered electrons

## Scattering by a crystal



## A

Unit cell (uc) Structure factor (F)

Polarization factor

## Diffracted Beam Intensity

- Structure factor
- Polarization factor
- Lorentz factor
- Multiplicity factor
- Temperature factor
- Absorption factor

$$
I_{C}(q)=m A L p K|F(q)|^{2}+I_{b}
$$

## The Structure Factor

$$
\begin{gathered}
F_{h k l}=\sum_{1}^{N} f_{n} e^{2 \pi i\left(h u_{n}+k v_{n}+l w_{n}\right)} \\
F_{h k l}=\frac{\text { amplitude scattered by all atoms in a unit cell }}{\text { amplitude scattered by a single electron }}
\end{gathered}
$$

- The structure factor contains the information regarding the types ( $f$ ) and locations ( $u, v, w$ ) of atoms within a unit cell.
- A comparison of the observed and calculated structure factors is a common goal of X-ray structural analysis.


## The Polarization Factor

- The polarization factor $p$ arises from the fact that an electron does not scatter along its direction of vibration
- In other directions electrons radiate with an intensity proportional to $(\sin \alpha)^{2}$ :


The polarization factor (assuming that the incident beam is unpolarized):

$$
p=\frac{1+\cos ^{2} 2 \theta}{2}
$$

## The Lorentz - Polarization Factor

- The Lorenz factor $L$ depends on the measurement technique used and, for the Diffractometer data obtained by the usual $\theta-2 \theta$ scans, it can be written as

$$
L=\frac{1}{\sin 2 \theta}
$$

- The combination of geometric corrections are lumped together into a single Lorentz-polarization (Lp) factor:

$$
L p=\frac{1+\cos ^{2} 2 \theta}{\sin 2 \theta}
$$

The effect of the Lp factor is to decrease the intensity at intermediate angles and increase the intensity in the forward and backwards directions

## The Temperature Factor

- As atoms vibrate about their equilibrium positions in a crystal, the electron density is spread out over a larger volume.
- This causes the atomic scattering factor to decrease with $\sin \theta / \lambda$ (or $|\mathbf{S}|=$ $4 \pi \sin \theta / \lambda$ ) more rapidly than it would normally.

The temperature factor is given by:

$$
\exp \left[-B \frac{\sin ^{2} \theta}{\lambda^{2}}\right]
$$

where the thermal factor $B$ is related to the mean square displacement of the atomic vibration:

$$
B=8 \pi^{2} \times \overline{u^{2}}
$$

This is incorporated into the atomic scattering factor:

$$
f \rightarrow f_{0} e^{-M} \Rightarrow f^{2} \sim e^{-2 M}
$$



## The Multiplicity Factor

- The multiplicity factor arises from the fact that in general there will be several sets of $h k l$-planes having different orientations in a crystal but with the same $d$ and $F^{2}$ values
- Evaluated by finding the number of variations in position and sign in $\pm h, \pm k$ and $\pm l$ and have planes with the same $d$ and $F^{2}$
- The value depends on $h k l$ and crystal symmetry
- For the highest cubic symmetry we have:

| $100, \overline{1} 00,010,0 \overline{1} 0,001,00 \overline{1}$ | $p_{100}=6$ |
| :--- | :--- |
| $110, \overline{1} 10,1 \overline{1} 0, \overline{1} \overline{1} 0,101,10 \overline{1}, \overline{1} 0 \overline{1}, \overline{1} 01,011,0 \overline{1} 1,01 \overline{1}, 0 \overline{1} \overline{1}$ | $p_{110}=12$ |
| $111,11 \overline{1}, 1 \overline{1} 1, \overline{1} 11,1 \overline{1} \overline{1}, \overline{1} 1 \overline{1}, \overline{1} \overline{1} 1, \overline{1} \overline{1} \overline{1}$ | $p_{111}=8$ |

## The Absorption Factor

- Angle-dependent absorption within the sample itself will modify the observed intensity

Absorption factor for thin films is given by:

$$
A=1-\exp \left(-\frac{2 \mu \tau}{\sin \theta}\right)
$$

where $\mu$ is the absorption coefficient, $\tau$ is the total thickness of the film

## Diffracted Beam Intensity

$$
\begin{gathered}
I \propto F_{h k l} F_{h k l}^{*}=\left|F_{h k l}\right|^{2} \\
I_{C}(q)=A p(L p) K|F(q)|^{2}+I_{b}
\end{gathered}
$$

where $K$ is the scaling factor, $I_{b}$ is the background intensity, $q=4 \sin \vartheta / \lambda$ is the scattering vector for $x$-rays of wavelength $\lambda$

$$
I_{C}(q)=\left[1-\exp \left(-\frac{2 \mu \tau}{\sin \theta}\right)\right] \frac{1+\cos ^{2} 2 \theta}{\sin 2 \theta} K|F(q)|^{2}+I_{b}
$$

## X-ray Diffraction Graph Of Pure Anthracene using ORIGIN



## X'Pert Highscore



## USING FULLPROF SUIT

FullProf has been mainly developed for Rietveld analysis (structure profile refinement) of neutron (nuclear and magnetic scattering) or X-ray powder diffraction data collected at constant or variable step in scattering angle $2 \theta$. The program can be also used as a Profile Matching without the knowledge of the structure.

## WhinPLOTR [CDIFX UMR6226 Rennes / ILL Grenoble]

## 

Format of the data file

## Format of data file:

$\bigcirc \mathrm{X}, \mathrm{Y}$ data $+\mathrm{INSTRM}=10$
$\bigcirc$ INSTRM=0: Free F.(Ti,step,Tf)
$\bigcirc$ INSTRM $=1$ : Old D1A
C INSTRM=3: D1B (ILL)
C INSTRM=4: Brookhaven(Synchr.)
C INSTRM=5: G4.1
C INSTRM=6: D2B/3T2/G4.2
C INSTRM=8: HRPT/DMC (PSI)
$\bigcirc$ INSTRM=9:RX (Socabim)
C INSTRM=11: Variable Time step

- GSAS data
$\bigcirc \mathrm{CPI}$ (Xrays)
C PANalytical formats
$\bigcirc$ INSTRM=14: ISIS normalized data
C 15. Rigaku RINT

OK
Cancel


## WinPLOTR

WinPLOTR is a software to plot and analyse powder diffraction patterns It can be used to plot raw or normalized data files coming from neutron And $x$-ray diffractometers as well as Rietveld files created by several Rietveld type refinement program.

WinPLOTR has also been developed to be preferential graphic interface for The Rietveld type FullProf program : edition of PCR input file ,plot Rietveld Type plots.

Win WinPLOTR [CDIFX UMR6226 Rennes / ILL Grenoble]
File Plot Options Points Selection $X$ space Calculations Rietveld plot options Text External applications Tools Help

## 




| - Information |  |  |
| :---: | :---: | :---: |
| Title, type of job: Rietveld, Integrated Intensities, Simulated Annealing, ... | General |  |
| Type of Patterns, profile, background, diffraction geometry, user-given scattering factors ... |  |  |
| Phase name, type of calculations (JBT), ATZ, contribution to patterns, symmetry, . |  |  |
| Number of cycles, relaxation factors, access to patterns and phases (atoms and profile) | Refi | ment |
| Constraints definitions, adding, deleting. modifying... |  | raints |
| Fixing range of parameters, distances, angles, magnetic moments and linear restraints | Box/R | straints |
| Output options for patterns and phases: Reflection lists, Fourier, distances, BVS |  |  |
| Profiles: 1 Phases: 1 | 6/3/2012 | 14:55:17 |

n-2-:
(손

The main window of EdPCR program contains a menu bar and a toolbar with the usual buttons. A brief information is obtained when you left the mouse on a button of the toolbar.

The information of the $P C R$ file is distributed in seven buttons:

## General

Define a general information as title, type of job: Rietveld, Profile Matching, Simulating Annealing.

## Patterns

Define patterns information: types of profile, background, geometry aspects...

## Phases

Define Phase information: Names, contribution to patterns, symmetry...

## Refinement

This button is the access to the most important part of EdPCR: editing structural and profile parameters and conditions of refinement. Atom positions, profile shape parameters, magnetic moments, micro structural parameters, etc ... are accessible through this button.

## Constraints

Define constrains for refinable parameters. You can modify, add and delete constrains relations easily by using mouse selection and clicks.

## Output

Access to the selection of output options for each phase and pattern. This allows selecting output files: Fourier, hkl-lists, files for other programs, etc.


## Patterns Information



| Profile Data Information: Pattern 1 |  |  |  |
| :---: | :---: | :---: | :---: |
| Data File / Format \| Refinement/Simulation | Pattern Calculation/Peak Shape | |  |  |  |
| Data File: pantra |  |  | Browse... |
| Format |  |  |  |
| $\sim \mathrm{D} 1 . \mathrm{A} / \mathrm{D} 2 \mathrm{~B}$ [0ld Format] | (*)Free Format [2thetal, step, 2T hetaF] | T Variable Time $\times$-ray Data |  |
| $\sim$ D1A/D2B/3T2/G42 | c Two Axis Instrument, G41 | $\bigcirc \times$ YSIGMA [ $\times$ MDATA] |  |
| $\bigcirc$ D1B [Old Format] | $\sim$ GSAS Format | $\sim \times$ Celerator [PANalytical] |  |
| $\cdots \mathrm{D} 1 \mathrm{~B} / \mathrm{D} 20$ | $\sim$ Socabim Software | $\sim$ ISIS multi-bank normalized |  |
| $\bigcirc \mathrm{D} 4 / \mathrm{D} 20 \mathrm{~L}$ | $\bigcirc$ Synchroton [Brookhaven] |  |  |
| $\sim$ DMC/HRPD [P.S.I.] | $\sim$ Synchroton [DBW/S Software] |  |  |



## The pseudo-Voigt function

The pseudo-Voigt function has been shown to provide a good approximation to most peaks.
The pseudo-Voigt can be given by the following equation:
$l(2 \theta)=I_{\text {hkl }}\left[\eta \mathbf{L}\left(2 \theta-2 \theta_{0}\right)+(1-\eta) \mathbf{G}\left(2 \theta-2 \theta_{0}\right)\right]$
where, respectively, $\mathbf{L}\left(2 \theta-2 \theta_{0}\right)$ and $\mathbf{G}\left(2 \theta-2 \theta_{0}\right)$ represent suitably normalised Lorentz and Gaussian functions,.



$\Rightarrow$ Solving L.S. equations..
$\Rightarrow$ 相riting results for cycle 1
$\Rightarrow$ R-Factors: 10.6 14.4
Chi2: 14.4 DKT-Stat.: 0.1862 Patt\#: 1
$\Rightarrow$ Expected : 3.79

1. 8698
$\Rightarrow$ Conventional Rietveld R-factors for Pattern: 1
$\Rightarrow$ Rp: 23.9 Rwp: 24.4 Rexp: 6.43
Chi2: 14.4
$\Rightarrow$ Global user-weigthed Chi2 (Bragg contrib.): 17.09
=> ---------> Pattern\#
1
$\Rightarrow$ Phase: 1
$\Rightarrow \quad$ Bragg R -factor: $\quad 0.2845 \mathrm{E}-03$
$\Rightarrow \quad \mathrm{RF}$-factor : 0.7207E-01
$\Rightarrow$ Normal end, final calculations and writing..
```
= CPU Time: 0.172 seconds
=>
    0.003 minutes
=> END Date:16/04/2012 Time => 11:57:09.484
```



$$
\begin{aligned}
& R_{F}=\frac{\left.\sum \mid I_{K}{ }^{\prime} \text { (obs' }\right)^{1 / 2}-I_{K}\left({ }^{\prime} \text { calc' }\right)^{1 / 2} \mid}{\sum I_{K}\left({ }^{\prime} \text { obs } s^{\prime}\right)^{1 / 2}} \\
& R_{B}=\frac{\sum \mid I_{K}\left({ }^{\prime} \text { obs' }\right)-I_{K}\left({ }^{\prime} \text { callc' }\right) \mid}{\sum I_{K}\left({ }^{\prime} o b s^{\prime}\right)} \\
& R_{p}=\frac{\sum\left|y_{i}(o b s)-y_{t}(c a l c)\right|}{\sum y_{i}(o b s)} \\
& R_{w p}=\left\{\frac{\sum w_{i}\left(y_{i}(o b s)-y_{t}(\text { calcs })\right)^{2}}{\sum w_{i}\left(y_{i}(o b s)\right)^{2}}\right\}^{1 / 2} \\
& \text { R-structure factor } \\
& \mathrm{R} \text {-Bragg factor } \\
& \text { R-pattern } \\
& \text { R-weighted pattern }
\end{aligned}
$$

The function that is minimised is the chi-square $\chi^{2}$ :

$$
\chi^{2}=\frac{\sum_{i} w_{i}^{*} *\left|Y_{\text {obs }}^{i}-Y_{\text {calc }}^{i}\right|^{2}}{N-P}
$$

where:
$\sum_{i}$ : summation over the N points of the fitted region.
$w_{i}:$ weighting factor $\left(w_{i}=\frac{1}{\sigma\left(Y_{o b s}^{i}\right)}\right)$
$Y_{o b s}^{i} \quad$ : observed counts
$Y_{\text {calc }}^{i}$ : calculated counts
P : number of refined parameters.

```
Cycle: 1
pured anthraceneb_INSTRMO.dat
```



Wavelength: 1.54000
2theta_min: 5.00000
2theta_max: 50.00000
Space group: P 21/m
Cell parameters: $8.54990 \quad 6.0100011 .17000$
Cell angles: 90.00000124 .6000090 .00000
> Number of reflexions: 101

|  | $h$ | $k$ | $l$ | mult | stl(A-1) | d_hkl(A) | 2theta(deg) |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 1 | 0 | 0 | 1 | 2 | 0.05438 | 9.19443 | 9.608 |
| 2 | -1 | 0 | 1 | 2 | 0.06014 | 8.31373 | 10.628 |
| 3 | 1 | 0 | 0 | 2 | 0.07105 | 7.03773 | 12.563 |
| 4 | -1 | 0 | 2 | 2 | 0.09001 | 5.55520 | 15.935 |
| 5 | 0 | 1 | 1 | 4 | 0.09939 | 5.03062 | 17.609 |
| 6 | -1 | 1 | 1 | 4 | 0.10266 | 4.87061 | 18.192 |
| 7 | 0 | 0 | 2 | 2 | 0.10876 | 4.59722 | 19.284 |
| 8 | 1 | 1 | 0 | 4 | 0.10940 | 4.57029 | 19.399 |
| 9 | 1 | 0 | 1 | 2 | 0.11132 | 4.49149 | 19.743 |
| 10 | -2 | 0 | 1 | 2 | 0.11988 | 4.17077 | 21.278 |
| 11 | -2 | 0 | 2 | 2 | 0.12028 | 4.15686 | 21.350 |
| 12 | -1 | 1 | 2 | 4 | 0.12257 | 4.07944 | 21.760 |
| 13 | -1 | 0 | 3 | 2 | 0.13601 | 3.67611 | 24.181 |
| 14 | 0 | 1 | 2 | 4 | 0.13693 | 3.65144 | 24.347 |
| 15 | 1 | 1 | 1 | 4 | 0.13897 | 3.59779 | 24.716 |
| 16 | 2 | 0 | 0 | 2 | 0.14209 | 3.51887 | 25.280 |
| 17 | -2 | 0 | 3 | 2 | 0.14310 | 3.49396 | 25.463 |
| 18 | -2 | 1 | 1 | 4 | 0.14592 | 3.42650 | 25.973 |
| 19 | -2 | 1 | 2 | 4 | 0.14625 | 3.41878 | 26.032 |
| 20 | -1 | 1 | 3 | 4 | 0.15944 | 3.13598 | 28.427 |
| 21 | 1 | 0 | 2 | 2 | 0.16016 | 3.12183 | 28.559 |
| 22 | 0 | 0 | 3 | 2 | 0.16314 | 3.06481 | 29.102 |
| 23 | 2 | 1 | 0 | 4 | 0.16465 | 3.03665 | 29.378 |

## $\otimes \Theta$ crystallab@ubuntu:~

File Edit View Search Terminal Help
crystallab@ubuntu:~\$ g++ d.cpp
crystallab@ubuntu:~\$ ./a.out
enter two theta value
9.608
d=9. 19431
crystallab@ubuntu:~\$ ./a.out
enter two theta value
10.628
d=8.31409
crystallab@ubuntu:~\$ ./a.out
enter two theta value
12.563
d=7. 03753
crystallab@ubuntu:~\$ ./a.out
enter two theta value
15.935
d=5.5551
crystallab@ubuntu:~\$ ./a.out
enter two theta value
17.609
d=5. 03059
crystallab@ubuntu:~\$ ./a.out
enter two theta value
18.192
$\mathrm{d}=4.87067$
crystallab@ubuntu:~\$ ./a.out
enter two theta value
19.399
$\mathrm{d}=4.57025$
crystallab@ubuntu:~\$ ./a.out
enter two theta value
19.743
d=4.49139
crystallab@ubuntu:~\$ ./a.out
enter two theta value

| 24 | -2 | 1 | 3 | 4 | 0.16553 | 3.02060 | 29.537 |
| :--- | ---: | ---: | :--- | :--- | :--- | :--- | :--- |
| 25 | 0 | 2 | 0 | 2 | 0.16639 | 3.00500 | 29.694 |
| 26 | 0 | 2 | 1 | 4 | 0.17505 | 2.85632 | 31.278 |
| 27 | -3 | 0 | 2 | 2 | 0.17587 | 2.84303 | 31.428 |
| 28 | -1 | 2 | 1 | 4 | 0.17692 | 2.82606 | 31.622 |
| 29 | 2 | 0 | 1 | 2 | 0.17867 | 2.79847 | 31.942 |
| 30 | -2 | 0 | 4 | 2 | 0.18001 | 2.77760 | 32.188 |
| 31 | -3 | 0 | 3 | 2 | 0.18042 | 2.77124 | 32.264 |
| 32 | 1 | 1 | 2 | 4 | 0.18048 | 2.77038 | 32.275 |
| 33 | 1 | 2 | 0 | 4 | 0.18092 | 2.76362 | 32.356 |
| 34 | 0 | 1 | 3 | 4 | 0.18313 | 2.73030 | 32.762 |
| 35 | -1 | 0 | 4 | 2 | 0.18658 | 2.67979 | 33.397 |
| 36 | -3 | 0 | 1 | 2 | 0.18767 | 2.66420 | 33.598 |
| 37 | -1 | 2 | 2 | 4 | 0.18917 | 2.64308 | 33.875 |
| 38 | -3 | 1 | 2 | 4 | 0.19455 | 2.56998 | 34.869 |
| 39 | 2 | 1 | 1 | 4 | 0.19709 | 2.53693 | 35.338 |
| 40 | -2 | 1 | 4 | 4 | 0.19831 | 2.52135 | 35.564 |
| 41 | -3 | 1 | 3 | 4 | 0.19868 | 2.51659 | 35.633 |
| 42 | 0 | 2 | 2 | 4 | 0.19878 | 2.51531 | 35.652 |
| 43 | 1 | 2 | 1 | 4 | 0.20019 | 2.49757 | 35.914 |
| 44 | -3 | 0 | 4 | 2 | 0.20023 | 2.49717 | 35.919 |
| 45 | -1 | 1 | 4 | 4 | 0.20429 | 2.44751 | 36.674 |
| 46 | -2 | 2 | 1 | 4 | 0.20508 | 2.43809 | 36.821 |
| 47 | -3 | 1 | 1 | 4 | 0.20529 | 2.43562 | 36.860 |
| 48 | -2 | 2 | 2 | 4 | 0.20531 | 2.43531 | 36.864 |
| 49 | 1 | 0 | 3 | 2 | 0.21172 | 2.36159 | 38.05 |
| 51 | -1 | 2 | 3 | 4 | 0.21491 | 2.32659 | 38.654 |
| 52 | -3 | 1 | 4 | 4 | 0.21682 | 2.30603 | 39.012 |
| 53 | 0 | 0 | 4 | 2 | 0.21752 | 2.29861 | 39.143 |
| 54 | 2 | 2 | 0 | 4 | 0.21880 | 2.28515 | 39.383 |
| 55 | -2 | 2 | 3 | 4 | 0.21946 | 2.27828 | 39.507 |
| 56 | 2 | 0 | 2 | 2 | 0.22264 | 2.24575 | 40.104 |
| 57 | -2 | 0 | 5 | 2 | 0.22415 | 2.23063 | 40.387 |
| 58 | 1 | 1 | 3 | 4 | 0.22748 | 2.19799 | 41.014 |
| 59 | 3 | 1 | 0 | 4 | 0.22880 | 2.18533 | 41.262 |
| 60 | 1 | 2 | 2 | 4 | 0.23095 | 2.16498 | 41.668 |
| 61 | -3 | 0 | 5 | 2 | 0.23139 | 2.16083 | 41.752 |
| 62 | 0 | 1 | 4 | 4 | 0.23289 | 2.14694 | 42.035 |
| 63 | 0 | 2 | 3 | 4 | 0.23303 | 2.14569 | 42.060 |
| 64 | -4 | 0 | 3 | 2 | 0.23393 | 2.13741 | 42.231 |
| 65 | 2 | 1 | 2 | 4 | 0.23768 | 2.10368 | 42.941 |
| 66 | -1 | 0 | 5 | 2 | 0.23883 | 2.09353 | 43.160 |
| 67 | -2 | 1 | 5 | 4 | 0.23909 | 2.09124 | 43.210 |
| 68 | -4 | 0 | 2 | 2 | 0.23976 | 2.08538 | 43.337 |
| 69 | -4 | 0 | 4 | 2 | 0.24057 | 2.07843 | 43.489 |
| 70 | -3 | 2 | 2 | 4 | 0.24211 | 2.06521 | 43.782 |


| 71 | 2 | 2 | 1 | 4 | 0.24415 | 2.04794 | 44.171 |
| ---: | ---: | ---: | ---: | ---: | :--- | :--- | :--- |
| 72 | -2 | 2 | 4 | 4 | 0.24513 | 2.03972 | 44.358 |
| 73 | -3 | 2 | 3 | 4 | 0.24544 | 2.03720 | 44.416 |
| 74 | -3 | 1 | 5 | 4 | 0.24589 | 2.03339 | 44.504 |
| 75 | 3 | 0 | 1 | 2 | 0.24809 | 2.01541 | 44.922 |
| 76 | -4 | 1 | 3 | 4 | 0.24828 | 2.01385 | 44.959 |
| 77 | -1 | 2 | 4 | 4 | 0.25000 | 2.00003 | 45.287 |
| 78 | -3 | 2 | 1 | 4 | 0.25081 | 1.99352 | 45.443 |
| 79 | -1 | 1 | 5 | 4 | 0.25291 | 1.97702 | 45.844 |
| 80 | -4 | 1 | 2 | 4 | 0.25379 | 1.97015 | 46.013 |
| 81 | -4 | 1 | 4 | 4 | 0.25455 | 1.96429 | 46.158 |
| 82 | 0 | 3 | 1 | 4 | 0.25544 | 1.95741 | 46.330 |
| 83 | -1 | 3 | 1 | 4 | 0.25673 | 1.94759 | 46.577 |
| 84 | -4 | 0 | 1 | 2 | 0.25723 | 1.94381 | 46.673 |
| 85 | -4 | 0 | 5 | 2 | 0.25872 | 1.93259 | 46.960 |
| 86 | 1 | 3 | 0 | 4 | 0.25950 | 1.92679 | 47.110 |
| 87 | -3 | 2 | 4 | 4 | 0.26034 | 1.92058 | 47.272 |
| 88 | 3 | 1 | 1 | 4 | 0.26167 | 1.91083 | 47.527 |
| 89 | 1 | 0 | 4 | 2 | 0.26441 | 1.89098 | 48.058 |
| 90 | -1 | 3 | 2 | 4 | 0.26532 | 1.88454 | 48.232 |
| 91 | 1 | 2 | 3 | 4 | 0.26928 | 1.85681 | 49.000 |
| 92 | -3 | 0 | 6 | 2 | 0.27002 | 1.85173 | 49.143 |
| 93 | -4 | 1 | 1 | 4 | 0.27035 | 1.84948 | 49.207 |
| 94 | 3 | 2 | 0 | 4 | 0.27039 | 1.84916 | 49.216 |
| 95 | 2 | 0 | 3 | 2 | 0.27043 | 1.84892 | 49.223 |
| 96 | -4 | 1 | 5 | 4 | 0.27177 | 1.83981 | 49.483 |
| 97 | 0 | 0 | 5 | 2 | 0.27190 | 1.83889 | 49.509 |
| 98 | -2 | 0 | 6 | 2 | 0.27203 | 1.83806 | 49.533 |
| 99 | 0 | 3 | 2 | 4 | 0.27225 | 1.83653 | 49.577 |
| 100 | 1 | 3 | 1 | 4 | 0.27328 | 1.82959 | 49.778 |
| 101 | 0 | 2 | 4 | 4 | 0.27386 | 1.82572 | 49.890 |

## Application of XRD

XRD is a nondestructive technique. Some of the uses of $x$-ray diffraction are;

1. Differentiation between crystalline and amorphous materials;
2. Determination of the structure of crystalline materials;
3. Determination of electron distribution within the atoms, and throughout the unit cell;
4. Determination of the orientation of single crystals;
5. Determination of the texture of polygrained materials;
6. Measurement of strain and small grain size.....etc

## Advantages and disadvantages of X-rays

## Advantages;

- X-ray is the cheapest, the most convenient and widely used method.
- X-rays are not absorbed very much by air, so the specimen need not be in an evacuated chamber.


## Disadvantage;

- They do not interact very strongly with lighter elements.

THANKYOU

